



Effect of annealing on creep behavior, hardness, roughness and electrochemical corrosion parameters of Co- Cr based alloy

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ABSTRACT

In the present work, the effect of annealing on creep behavior, hardness, maximum shear stress, roughness and electrochemical corrosion parameters of commercial $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ (A= C, Si, Fe, and Mn) dental alloy from Travagliato (BS) - Italy have been studied and analyzed. Creep behavior was studied by indentation and stress exponent was determined by Mulheam-Tabor method. The results show that, Vickers hardness of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy decreased but roughness parameters varied after annealing for two hours at 700, 800 and 900 °C. Also the corrosion resistance in 0.5M HCl of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy is increased but the corrosion rate with 0.5M HCl is decreased after annealing compared to normal alloy.

Key words: - creep, hardness, roughness, corrosion parameters

1. INTRODUCTION

Cobalt-chromium alloys are widely used in prosthetic dentistry for fabrication of removable partial dentures and some fixed prosthetic appliances. Advantages of these dental alloys for casting prosthetic appliances are their low weight and good mechanical properties, such as great hardness, strength, resistance to tarnish and high temperatures and also resistance to corrosion. Cobalt–chromium based alloys have excellent corrosion resistance [1–5]. These alloys are mainly used for the fabrication of removable partial dentures because of their outstanding mechanical properties [6]. Since the beginning of the 20th century, Co–Cr–Mo alloys had used as cutting tools, valves of safety, turbines and strategic materials. They are also employed in bolts, screws and, in particular, in orthodontic prostheses [7-9]. The aim of this work was to study the effect of annealing on electrochemical corrosion parameters, indentation creep, hardness and roughness of commercial $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ dental alloy.

2. MATERIAL AND METHODS

The specimen used in the present work is a commercial $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ (A= C, Si, Fe, Mn) alloy from Travagliato (BS) – Italy were annealed at 700, 800 and 900 °C in furnace and leave in for two hours. The specimens were prepared in convenient shape for all tests such as roughness, Vickers microhardness and electrochemical corrosion behavior. Microhardness test of used specimens were conducted using a digital Vickers microhardness tester, (Model FM–7, Tokyo, Japan), applying a load of 100 gf for 5 seconds via a Vickers diamond pyramid. The roughness of used samples were measured by using surface roughness measurements device (S.J 201.P). For measuring corrosion parameters, the potentiodynamic current versus potential curves were recorded by changing the electrode potential automatically from –1500 to 500mV at a scan rate of 5 mV/s¹ using Voltage lab PGZ 100 (Germany PC 3–300) and a computer with Volta Master 4 software (Germany frame work version 7.08) for calculations. For indentation creep, Vickers microhardness tester at different load and time was used. There exists a linear relationship between indentation time and hardness for all

conditions. The slope of the resultant lines according Mulheam-Tabor method is $-\left(n + \frac{1}{2}\right)$ where n is the stress exponent [10].

3. RESULTS AND DISCUSSION

3.1 Creep behavior

Creep behavior of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy was studied by indentation method performed at room temperature. Figure (1a) shows the indentation creep data where the indentation length is plotted versus the indentation time at constant loads (100, 200 and 300 g). The indentation length increased with the loading time and the applied load increased. The curves shown in Figure (1a) consist of two stages similar to an ordinary creep curve. The first stage of the curve records an increase in the indentation length with loading time followed by a steady-state region where indentation sizes increase linearly with time. Fracture of the specimen dose not occurs and it is not possible to record a third stage of the curve as opposed to what happens in an ordinary creep test because the hardness test is a compression test.

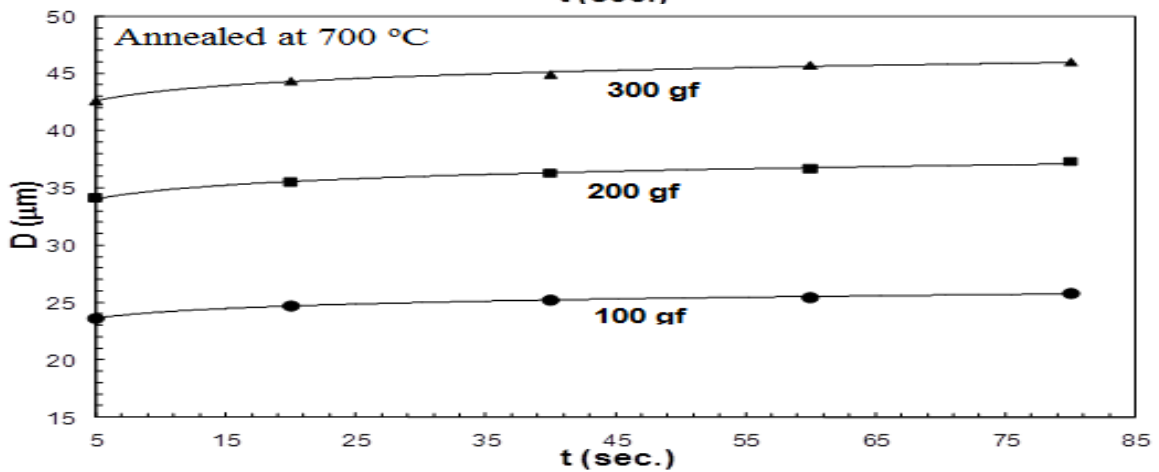
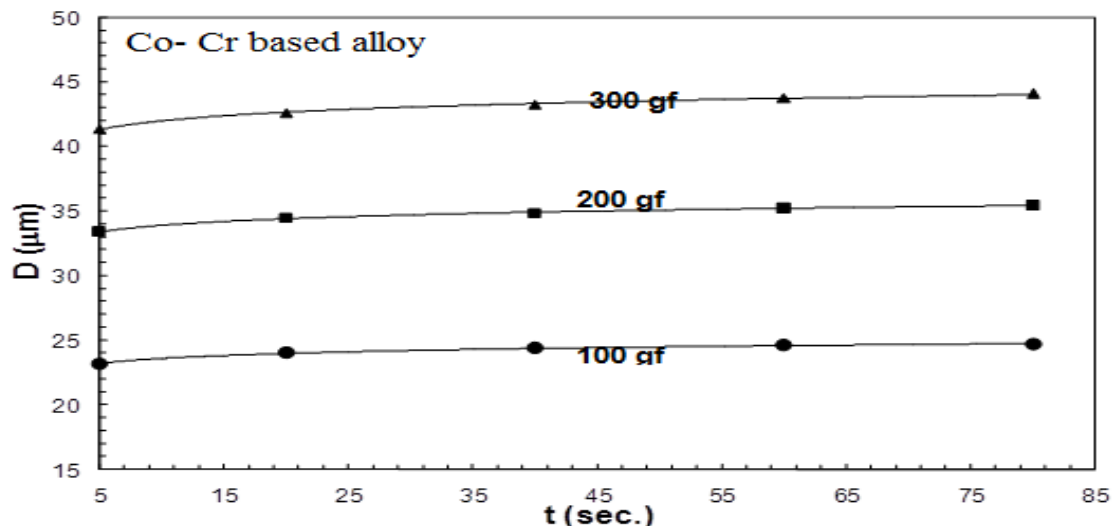
In the Mulheam-Tabor method, Figure (1b), Vickers hardness number of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy before and after annealing for different temperature at two hours is plotted versus indentation time on log-log scale for the indentation data. The stress exponent values of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy before and after annealing are given in Table (1). The exponent values are in the range of 21.73 to 13.52 depending on annealing temperature of used alloy. The change in stress exponent values are attributable to microstructural features, (changing in matrix alloy such as lattice parameters, solid



solution, size and distribution of strengthening phases and intermetallic phases) and that is agree with pervious results [11-13]. The grain boundary decreased with increasing the grain size, which dislocation movement affecting on stress exponent values [14].

Table 1:- Stress exponent of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{Al}_{0.5}$ alloy before and after annealing

Alloys	Stress exponent
Based alloy	21.73
700 °C	16.84
800 °C	14.65
900 °C	13.52



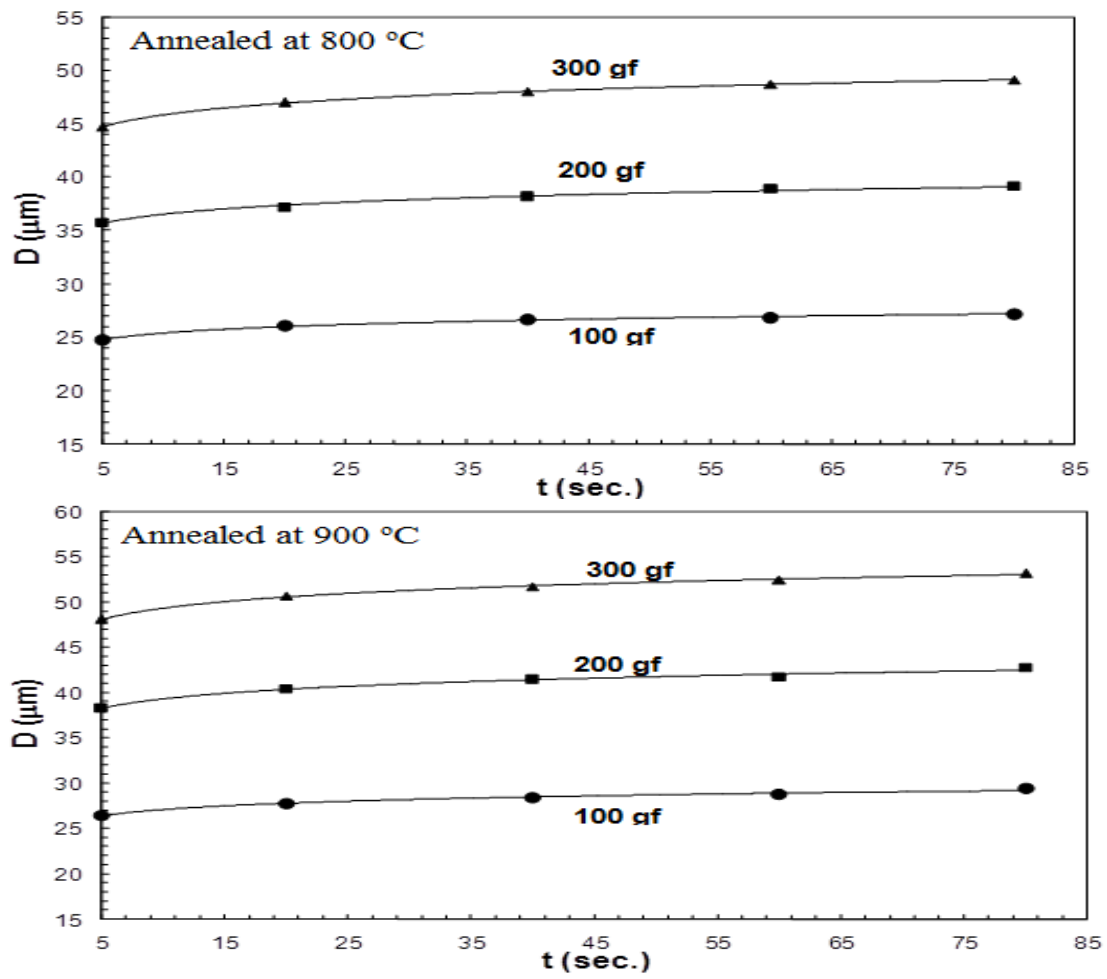


Fig. 1a:-the indentation length is plotted versus the indentation time

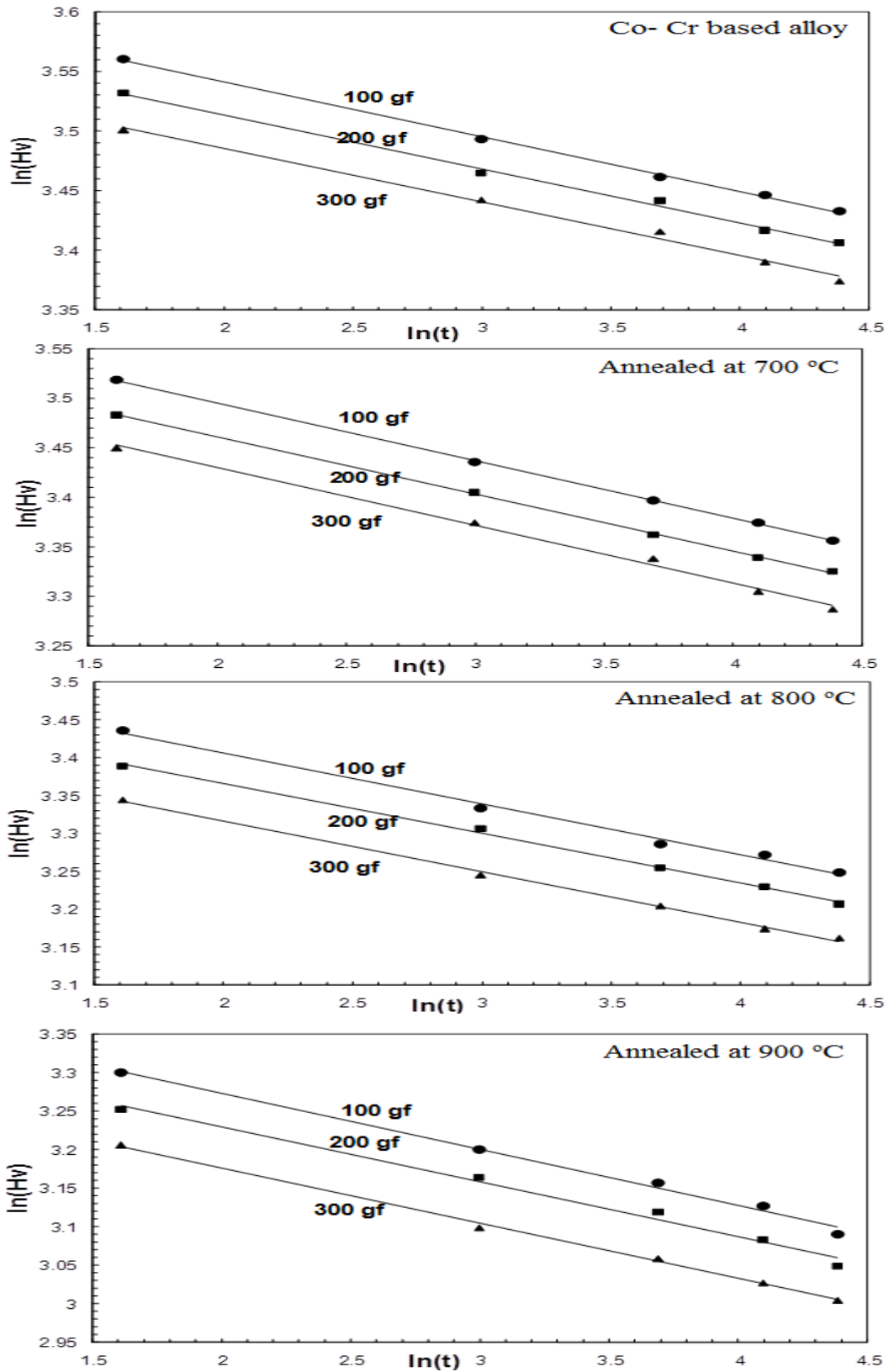


Fig. 1b:- Vickers hardness versus indentation time on log-log scale



3.2 Vickers hardness

The microhardness number was conducted using a digital Vickers microhardness tester, applying a load of 100 g for 5 s, for $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy. Vickers hardness value of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy before and after annealing at 700, 800 and 900 °C for two hours is listed in Table 2. Vickers hardness value of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy decreased after annealing. That is because annealing caused a softening in matrix alloy with eliminate the dislocations which attributed to a homogenization in composition. Also grain growth will occur with changing a bonding strength of the alloy as a result of heat treatment which reducing its hardness and that is agree with pervious results [15].

The maximum shear stress (μ_m) value of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy before and after annealing was calculated [16] and then presented in Table 2.

Table 2:- Vickers hardness and maximum shear stress of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy before and after annealing

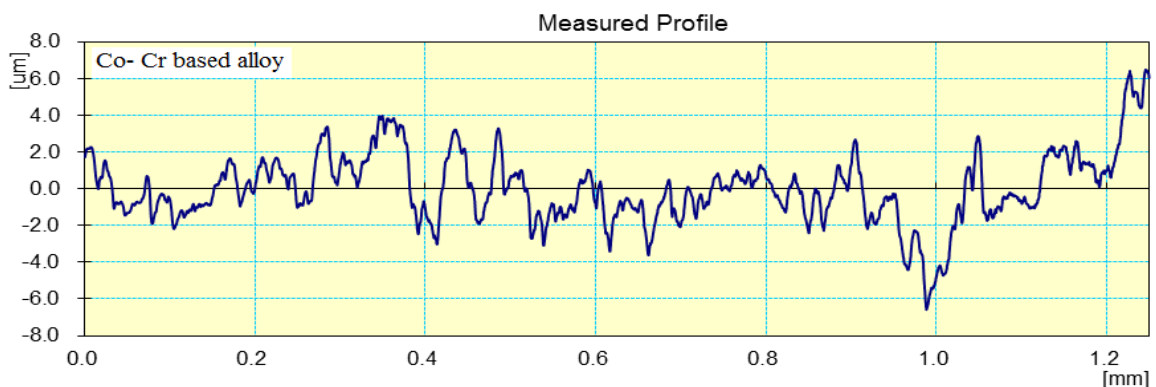
Alloys	H_v kg/mm^2	μ_m kg/mm^2
Based alloy	345.1±23.21	113.88
700 °C	330.9±23.64	109.2
800 °C	304.8±24.77	100.58
900 °C	265.9±26.42	87.75

3.3 Roughness

The roughness profiles of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy before and after annealing at 700, 800 and 900 °C for two hours are shown in Figure 2. The average surface roughness parameter Ra along the total sliding distance and other roughness parameters of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ alloy before and after annealing are listed in Table 3. The results show, the annealing caused variation in average surface roughness parameter Ra and other roughness parameters. That is mean, numbers of formed cracks and/or pits during annealing are varied

Table 3:- roughness parameters of $Co_{64}Cr_{29}Mo_{6.5}A_{0.5}$ before and after annealing

Roughness parameters	Samples			
	base	700 °C	800 °C	900 °C
Ra μm	1.06	1.12	1.08	1.07
Rz μm	5.84	6.74	5.58	5.84
Rq μm	1.35	1.43	1.35	1.34
Rt μm	8.84	10.6	7.37	7.04
Rp μm	2.73	3.48	2.95	2.41



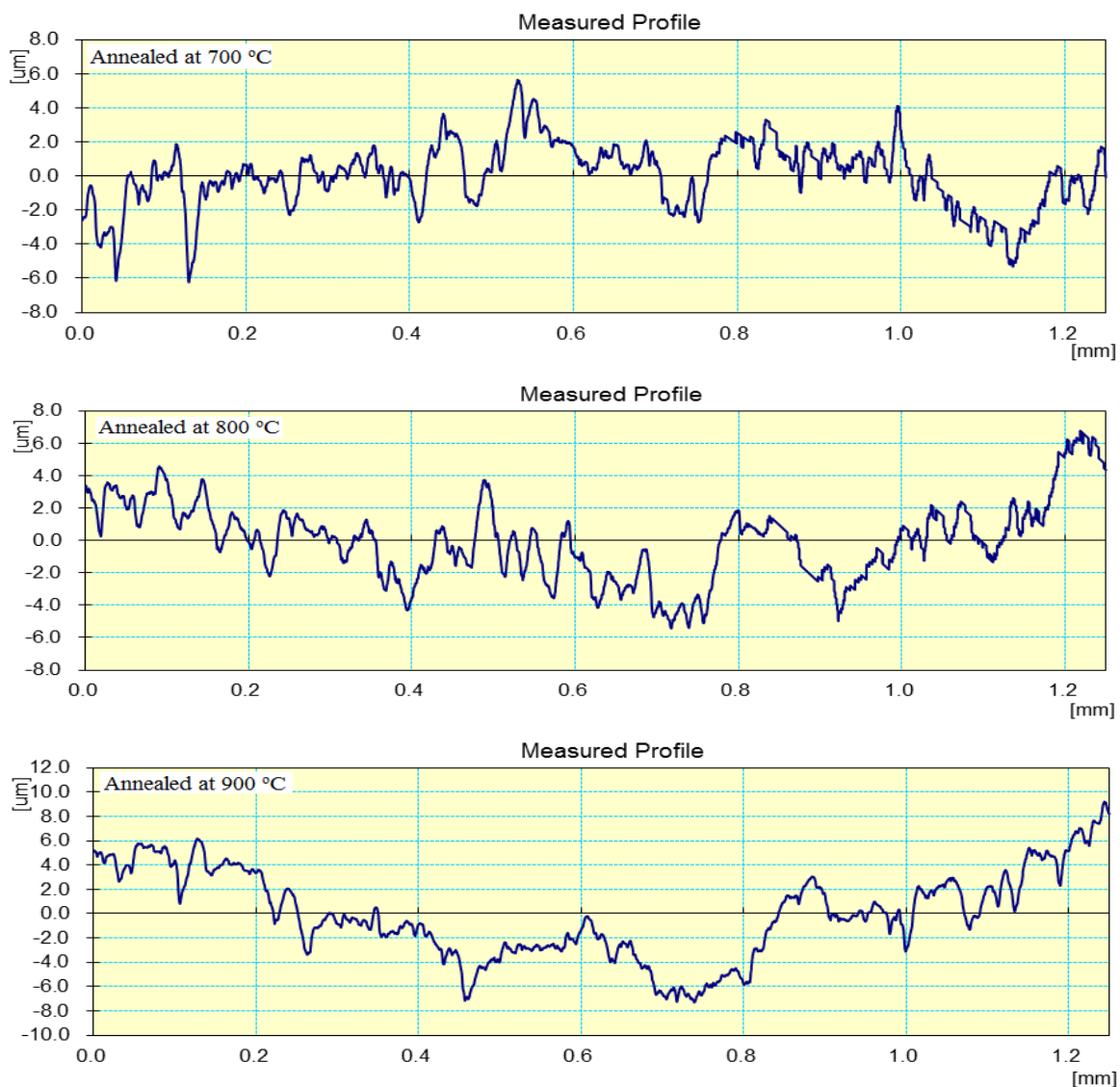


Fig. 2:- roughness profiles of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy before and after annealing

3.4 Electrochemical corrosion behavior

Table 4 presented the corrosion potential (E_{Corr}), corrosion current (I_{Corr}), corrosion resistance (R_p) and corrosion rate (C.R) of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy before and after annealing. From these results, it is clear that the corrosion resistance in 0.5M HCl of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy is increased but the corrosion rate with 0.5M HCl is decreased after annealing. That is meant annealing caused heterogeneous microstructure with affected on microsegregation and reactivity of Co and other atoms with HCl solution

Table 4:- corrosion parameters of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ before and after annealing

Alloys	E_{corr} mV	I_{corr} $\mu\text{A}/\text{cm}^2$	C. R mm/yr	R_p Ohm cm^2
Based alloy	-341.3	71.9	0.836	0.69 E+03
700 °C	-333.3	39.8	0.462	1.1 E+03
800 °C	-326.4	42.43	0.494	1.01E+03
900 °C	-334	54.5	0.634	0.89E+03



CONCLUSION

The average surface roughness parameter Ra of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy increased after annealing at different temperature for two hours

- 1- Stress exponent and Vickers hardness values of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy decreased after annealing for two hours
- 2- Roughness parameters of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ alloy varied after annealing.
- 3- The corrosion resistance of $\text{Co}_{64}\text{Cr}_{29}\text{Mo}_{6.5}\text{A}_{0.5}$ in 0.5M HCl increased but the corrosion rate with 0.5M HCl decreased after annealing.

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